

Electrocaloric effect in La-doped BNT-6BT relaxor ferroelectric ceramics

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Abstract: Relaxor $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0, 0.03, 0.06, 0.09$) ceramics (La-doped BNT-6BT hereafter) with composition close to the morphotropic phase boundary (MPB) were successfully prepared by using the conventional solid state reaction method. All samples present almost a pure perovskite phase with coexistence of tetragonal and rhombohedral. With the increase of La doping content, the degree of the dielectric relaxor dispersion around the dielectric peak which is close to the room temperature increases, and also the transition temperature of ferroelectric-to-relaxor (T_{F-R}) shifts 120K towards a lower temperature at $x=0.09$. The maximum value of the temperature change (ΔT) of the electrocaloric (EC) effect decreases sharply from 1.1 K at $x = 0$ to 0.064 K at $x = 0.09$. A large positive EC effect (maximum $\Delta T \sim 0.44$ K) in a broad temperature range (~ 90 K) close to room temperature is achieved at $x = 0.03$, indicating that it is a promising lead-free material for application in solid state cooling system. Moreover, it is found that the Maxwell relationship can be well used to assess the EC effects of the La-doped BNT-6BT

ceramics when the operating temperature is higher than that of the T_{F-R} , indicating that these relaxor ceramics would perform as an ergodic.

Keywords: Relaxor ferroelectric, MPB, Lead-free, Broad temperature range

1. Introduction

The *EC* effect has been of much interest in cooling systems. It can be designed to substitute conventional mechanical vapor compression cycle cooling devices such as refrigerators and air conditioners which release a large number of greenhouse gases during their operation [1]. In the past, the bottleneck for the development of *EC* cooling technologies was that only small temperature (ΔT) and entropy changes (ΔS) could be induced in bulk materials by virtue of the low breakdown electric field [2]. Fortunately, a giant *EC* effect in a $\text{PbZr}_{0.95}\text{Ti}_{0.05}\text{O}_3$ (PZT) thin film with a large ΔT of 12 K was first reported by Mischenko et al. in 2006, which revived great interest in this field [3]. Further research activities lead to even larger ΔT and ΔS in $\text{Pb}_{0.8}\text{Ba}_{0.2}\text{ZrO}_3$ (PBT) [1], $\text{Pb}(\text{Sc}_{0.5}\text{Ta}_{0.5})\text{O}_3$ (PST) [4], $\text{Pb}(\text{Mg}_{1/3}\text{Nb}_{2/3})_{0.65}\text{Ti}_{0.35}\text{O}_3$ (PMN-PT) [5] and $(\text{Pb}_{0.88}\text{La}_{0.08})(\text{Zr}_{0.65}\text{Ti}_{0.35})\text{O}_3$ (PLZT) [6], etc. **Similar** ΔT values were also reported in copolymers (P(VDF-TrFE)) [7, 8] and terpolymers (P(VDF-TrFE-CFE)) [9, 10]. However, when the electrocaloric coefficient ($\Delta T/\Delta E$) is taken into consideration, it becomes clear that inorganic oxides are much better *EC* materials [11] compared to polymers. In recent years, in order to respond to the European "RoHS/WEEE" environmental and health protection directive, lead-free ferroelectric materials have been extensively researched and rapidly developed in *EC*

effect, energy storage and piezoelectricity, etc.

Lead-free solid solutions based on the perovskites, such as BaTiO_3 (BT), $\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ (BNT), $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ (KNN) and $\text{Bi}_{0.5}\text{K}_{0.5}\text{TiO}_3$ (BKT) are promising environmentally friendly ferroelectrics that have been intensively researched [12-15]. Among them, $0.94(\text{Bi}_{1/2}\text{Na}_{1/2})\text{TiO}_3$ - 0.06BaTiO_3 (BNT-BT) ceramics have been widely studied to replace the lead-based ceramics. Numerous researchers have focused on compositional modifications to improve the performance of the BNT-BT system [16, 17]. It was found that the materials with the composition near the MPB coexisting with rhombohedral and tetragonal phase structures can result in good performance on *EC* effects and typical relaxation properties. For example, $0.935\text{Bi}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ - 0.065BaTiO_3 - SrTiO_3 (BNBST) [18], $\text{Ba}_{0.5}\text{Na}_{0.5}\text{TiO}_3$ - $x\text{BaTiO}_3$ (BNT- x BT) [19] and $(\text{Na}_{0.5}\text{Bi}_{0.5})_{1-x}\text{Ba}_x\text{TiO}_3$ (BNBT) [20, 21], etc.

Apart from the perovskites above mentioned, the *EC* effect of other perovskite materials, such as $\text{Pb}(\text{Sc}_{0.5}\text{Ta}_{0.5})\text{O}_3$ (PST) [22], BaTiO_3 (BT) [23] and $(1-x)(\text{Na}_{0.5}\text{Bi}_{0.5})\text{TiO}_3$ - $x\text{KNbO}_3$ [24], etc., have also been considered. Along with large ΔT and ΔS , a broad operable temperature range is required in practical cooling application as well. La has been used to modify several archetypal perovskites, such as $\text{Pb}(\text{Zr,Ti})\text{O}_3$ and $\text{Ba}_{1/2}\text{Na}_{1/2}\text{TiO}_3$ [25-27]. Its isovalent and aliovalent doping has shown to be a reliable strategy to tailor the dielectric, pyroelectric and relaxor properties of the BNT based ceramics. For instance, $(\text{Bi}_{0.94-x}\text{La}_x\text{Na}_{0.94})_{0.5}\text{Ba}_{0.06}\text{TiO}_3$ [25-xxx], La-doped $\text{Bi}_{1/2}(\text{Na}_{0.82}\text{K}_{0.18})_{1/2}\text{TiO}_3$ [28] and La-doped

$\text{Bi}_{1/2}\text{Na}_{1/2}\text{TiO}_3\text{-BaTiO}_3\text{-SrZrO}_3$ [29], etc.

In this study, an A-site La doped $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0, 0.03, 0.06, 0.09$) system is designed to obtain a wider operating temperature range. It can be concluded that La doping makes the dielectric relaxor dispersion around the dielectric peak near room temperature more diffuse and reduces the T_{F-R} by nearly 120 K at $x = 0.09$. The maximum value of ΔT decreases sharply from 1.1 K at $x = 0$ to 0.064 K at $x = 0.09$. The La-doped BNT-6BT ceramics at $x = 0.03$ presents a large positive EC effect (maximum $\Delta T \sim 0.44$ K) in a broad temperature range (~ 90 K) at near room temperature, which makes it more practical for solid state EC effect in the future applications. Moreover, it is found that the Maxwell relationship could be well applied to assess the EC effects of the La-doped BNT-BT ceramics when the value of the operating temperature is higher than that of the T_{F-R} where relaxor ceramics perform at the almost ergodic state.

2. Experimental process

La-doped BNT-6BT ceramics were prepared by conventional solid state reaction method. The starting materials were BaCO_3 (purity $\geq 99\%$), La_2O_3 (purity $\geq 99.5\%$), Na_2CO_3 (purity $\geq 99.8\%$), Bi_2O_3 (purity $\geq 99\%$) and TiO_2 (purity $\geq 99\%$). The raw materials were weighed at a stoichiometric ratio and then ball milled in a nylon tank with the zirconia balls as media in ethanol for 4 hours. The dried slurries were calcined at 900°C for 6 hours and ball milled again for 12 hours to dissociate the agglomerates. After further drying, the ceramic powders were pressed in a steel mold

with 300 MPa for 60 s into green disks with 10 mm in diameter and 1 mm in thickness. All the pellets were then sintered at 1150 °C for 6 hours with a heating rate of 3 °C/min, and then cooled to room temperature naturally.

The density of the samples was measured by the Archimedeian drainage method. The phase structure of ceramics was analyzed by an X-ray diffractometer (XRD, X'Pert PRO MPD, Philips, Eindhoven, Netherlands). The microstructure of the ceramics was observed by a field emission scanning electron microscopy (JEOL-6700F, Japan Electron Co, Tokyo, Japan). The dielectric response was measured using a precision impedance analyzer (4294A, Agilent, CA, USA). The temperature control was performed using a temperature controller (TP94, Linkam, Surrey, UK) with an excitation voltage of 500 mV/mm, with the frequency and temperature acquisition ranging from 100 Hz to 1 MHz and 213 K to 720 K with a ramp rate of 3 K/min respectively. The hysteresis loops (P - E) were tested using a ferroelectric analyzer (TF-2000, AixACCT, Aachen, Germany) at a frequency of 1 Hz.

3. Results and analysis

3.1 Structure and morphology

Figure 1 a) shows the XRD patterns of sintered La-doped BNT-6BT ceramics. From the XRD patterns, it can be seen that after 6 hours sintering at 1150°C all the components were pure phase perovskite structure without pyrochlore phase coexisting. (001) and (002) diffraction peaks of La doped BNT-6BT ceramics gradually shifted to the low-angle direction with the increase of La content which indicates the increase of

its pseudocubic lattice parameter a . The structure of BNT system is rhombohedral and that of BT is tetragonal at room temperature [30]. A rhombohedral-tetragonal MPB exists in their solid solutions near the BNT-6BT composition [31]. From **Figure 1 b)**, it can be seen that both reflections at about 40.2° and 46.6° exhibit the feature of peaks splitting, which could be assigned to the (003) and (021) reflections of the rhombohedral phase and (002) and (200) reflections of the tetragonal phase, respectively. This result reveals that the La-doped BNT-6BT ceramics remain in the coexistence of the rhombohedral-tetragonal phases as in BNT-6BT ceramics.

Figure 2 shows the SEM images of La-doped BNT-6BT ceramics after the thermal etching (1100 °C insulation for 6 h). The SEM images show that the La-doped BNT-6BT ceramics have a dense microstructure with nearly uniform grain size. With the increase of La content, the plate-like grains first increase and then decrease, and the the largest plate-like number appeared at $x = 0.06$. The average size of plate-like grains of La-doped BNT-6BT ceramic samples remained roughly the same with an average size of 3 microns, indicating that the grain size might not play an important role so that it could affect the electrocaloric properties of La-doped BNT-6BT ceramics.

3.2 Dielectric properties

Figures 3 depicts the temperature dependence of the dielectric constant ($\varepsilon(T)$) and the dielectric loss ($\tan \delta(T)$) of the La-doped BNT-6BT ceramics at different frequencies. The $\varepsilon(T)$ curves of all the samples show a strong frequency dispersion in the

low-temperature region and a weak frequency dispersion in the high-temperature region. The study results of Wook et al. [20] show that there may be two independent relaxation processes in the different temperature ranges. The dielectric anomaly at the temperature near the dielectric permittivity maximum (T_m) corresponding to high-temperature dielectric constant peak is related to a relaxation of tetragonal polar nanoregions (PNRs) emerged from rhombohedral PNRs and the other dielectric anomaly at lower temperatures is due to thermal evolutions of discrete PNRs. Recent studies demonstrated that the dielectric anomaly at lower temperatures should be referred to as ferroelectric-to-relaxor transition temperature (T_{F-R}) rather than depolarization temperature (T_d) [32, 33]. The two dielectric anomalies on the dielectric $\varepsilon(T)$ curves are always observed in the BNT-based ceramics. Three consecutive events are expected to happen during heating. The initially existing mixture of polar R3c rhombohedral clusters and a weakly polar or ferroelectric P4bm tetragonal matrix undergo a thermal evolution resulting in the first broad dielectric maximum, followed by a transition of R3c to P4bm PNRs, and finally all the P4bm symmetries undergo a thermal evolution giving rise to the second frequency-dispersive dielectric maximum [20]. The nanodomains in the P4bm phase are actually uncompensated antiferroelectric, or weakly polar ferroelectric domains [34]. As a result, the dielectric behavior of the P4bm relaxor antiferroelectric phase is similar to that of a relaxor ferroelectric in many aspects. Ma and Tan have proposed a “relaxor antiferroelectric” term to describe the dielectric behavior of the P4bm phase for its strong frequency dispersion in their previous works [35, 36].

With the increase of La content, the dielectric diffusion peak of T_{F-R} shifts to the low temperature direction, while the peak of T_m shifts to the high temperature direction. Broad dielectric constant peak could be well observed and indicates a diffused phase transition behavior. This might be originated from compositionally induced ferroelectric to diffused phase transition, and multiple complexes in the A-site of perovskite compounds is another reason [37, 38]. At the temperature higher than T_m , the dielectric loss of La-doped BNT-6BT increases abnormally with the temperature increasing, which could be attributed to high conduction property in the high temperature region [30].

In Lorentz function $\varepsilon_A/\varepsilon=1+(T-T_A)^2/2(\delta_A)^2$ [39, 40], $\varepsilon(T)$ curves could be decomposed into two parts due to the contributions of high and low temperatures respectively. Dot lines in **Figure 3** show the Lorentz decomposition of the $\varepsilon(T)$ curves at 100 Hz and **Table 1** shows the fitting of the Lorentz parameter for $\varepsilon(T)$ curves. As can be seen from the **Table 1**, the δ_A of La-doped BNT-6BT ceramics gradually increases from 80 of the composition $x = 0$ to 118 of the composition $x = 0.09$, indicating that the addition of La improves the relaxation dispersion of the system on the other hand.

3.3 Electrocaloric Effect

In order to study the electrocaloric effect of La-doped BNT-6BT, the hysteresis loops (P - E) during temperature rising at the interval of 30 K were measured as shown in **Figures 4 a)-d)**. At $x = 0$, there is an evident double-like hysteresis loops above 330 K accompanying phase transformation from ferroelectric R3c phase to antiferroelectric P4bm phase. One abnormal behavior is that the values of maximum polarization (P_{max})

increase with temperature, and then decrease, as depicted in **Figure 4 a)**. This abnormal behavior is also observed by Bai et al.[23], which is caused by the phase transition induced by the external field from ferroelectric (FE) to antiferroelectric (AFE). At $x = 0.03, 0.06$ and 0.09 , the P - E loops become much slimmer compared to the unmodified BNT-6BT, indicating that the La substitution causes the lattice distortion and disrupts the dominant ferroelectric long-range-order leading to a decrease in the remnant polarization and maximum polarization [41]. However, the ergodic phase dominates at higher La contents ($x = 0.06, 0.09$) as confirmed by the drastic reduction in the maximum polarization, resulting in the delay transformation of relaxor to ferroelectric phase [42, 43].

Under the condition of reversible adiabatic approximation, the Maxwell relational expression $(\partial P/\partial T)_E = (\partial S/\partial E)_T$ is assumed to be valid. The EC effect of La doped BNT-6BT ceramics can be obtained from equations (1) and (2) [3, 7].

$$\Delta T = -\frac{1}{\rho} \int_{E_1}^{E_2} \frac{T}{C} \left(\frac{\partial P}{\partial T} \right)_E dE \quad (1)$$

$$\Delta S = -\frac{1}{\rho} \int_{E_1}^{E_2} \left(\frac{\partial P}{\partial T} \right)_E dE \quad (2)$$

In equations (1) and (2), T is the operating temperature, P the maximum polarization at the applied electric field E , and E_1 and E_2 the initial and final applied electric field respectively. The density of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics are: 5.1638 g/cm^3 ($x = 0$), 5.589 g/cm^3 ($x = 0.03$), 5.574 g/cm^3 ($x = 0.06$), 5.1736 g/cm^3 ($x = 0.09$). For simplicity, the heat capacity is assumed to be $0.45 \text{ J/(K}\cdot\text{kg)}$. The pyroelectric coefficient $((\partial P/\partial T)_E)$ could be obtained from the temperature dependence of

polarization ($P(T)$), which are extracted from the upper branches of the P - E loops in $E > 0$. **Figure 5 a) - d)** are given for temperature dependence of pyroelectric coefficient ($(\partial P/\partial T)_E(T)$) curves of La-doped BNT-6BT ceramics at selected electric field. The illustrations correspond to $P(T)$ curves under selected electric fields, where the solid curves are the 6th order polynomial fitting of the $P(T)$ curves. It shows that the absolute value of $(\partial P/\partial T)_E$ decreases with the increase of La content. Overall, the $(\partial P/\partial T)_E$ value of the composition at $x = 0.03$ is not deviated far from that of unmodified BNT-6BT, but the values of $(\partial P/\partial T)_E$ reduce significantly for the composition at $x = 0.06, 0.09$, anticipating a larger ECE for composition at $x = 0.03$ compared to other modified BNT-6BT ceramics. Besides, it can be found that the electric field has little influence on the $(\partial P/\partial T)_E(T)$ curves, which promises a larger ΔT when it comes to the La-doped BNT-6BT thin film.

Figure 6 a) - d) show the temperature dependence of $\Delta T(T)$ of La-doped BNT-6BT ceramics under different electric fields. It can be found that the $\Delta T(T)$ curves with different electric fields are intertwined for all components in the lower temperature region, indicating a strong or middle interaction between the domains respectively. While in the higher temperature region, the $\Delta T(T)$ curves increase regularly with the electric field and show a good application of Maxwell relationship, implying a state similar to the ergodic state with weak interaction between domains. Moreover, seen from **Figure 6** and **Figure 3**, an interesting phenomenon is found that the La-doped BNT-6BT ceramics would transform to an ergodic state with weak interaction between domains at the temperature close to T_{F-R} . For the composition at $x = 0$ in the

vicinity of the morphotropic phase boundary, its ΔT appears negative value when the temperature is below 355 K. The negative *EC* effect has been found in other materials, it is attributed to the non-collinearity of the applied electric field and the polarization direction after the phase transition occurs from the rhombohedral to the tetragonal [44]. The temperature where the negative *EC* effect vanished corresponds well with the pinching temperature of the *P-E* loops shown in **Figures 4 a)**, which could be attributed to the thermal evolution from polar R3c rhombohedral clusters to antiferroelectric P4bm tetragonal phase. The original ferroelectric system with strong interaction between domains is disrupted by the increase of antiferroelectric P4bm phase, resulting in the reduction of the interaction between domains and the appearance of positive *EC* effect. For the La doping components, it can be seen that there are two ΔT peaks in the temperature dependent curves, which are related to an increase in relaxation dispersion [22, 45-48]. La doping could make the transition happen easily in the low temperature and induce a decrease of T_{F-R} for about 120 K at $x = 0.09$, which promises a large *EC* effect near room temperature. The maximum value of the temperature change (ΔT) of the *EC* effect decreases sharply from 1.1 K at $x = 0$ to 0.064 K at $x = 0.09$. On the whole, the La-doped ceramic with the composition at $x = 0.03$ is ideal, and La doping has reduced the T_{F-R} of the ceramics from 456.9 K to 382.2 K for nearly 80 K, and a large positive *EC* effect (maximum $\Delta T \sim 0.44$ K) in a broad temperature range from 334.26 K to 427.59 K at the electric field of 51.97 KV/cm. Its broader temperature range near room temperature and larger ΔT make it more likely to meet the actual needs. **Figure 7 a) - d)** show the

temperature dependence of entropy variation ($\Delta S(T)$) of La-doped BNT-6BT ceramics under different electric fields. It can be found that the variation of ΔS is basically the same as that of ΔT , so only the change relation diagrams were given here.

4. Conclusion

A broad operable temperature range near room temperature and large ΔT have been achieved in relaxor $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0.03$) ceramics prepared using the conventional solid state reaction method. The composition of the material was near the morphotropic phase boundary with the rhombohedral and tetragonal phase coexisting. It can be concluded that the La doping could make the dielectric relaxation dispersion around the dielectric peak near the room temperature more diffuse and decrease the T_{F-R} for nearly 120 K at $x = 0.09$, resulting in a broader operational temperature range near room temperature. A large positive EC effect (maximum $\Delta T \sim 0.44$ K) in a broad temperature range (~ 90 K) close to room temperature is achieved at $x = 0.03$. The good EC performance of the $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0.03$) relaxor ferroelectric ceramic may make it a promising lead free material for future applications in the cooling system. Moreover, it is found that the Maxwell relationship can be well used to assess the EC effects of the La-doped BNT-6BT ceramics when the value of the operating temperature is higher than that of the T_{F-R} , indicating that these relaxor ceramics would perform as an ergodic.

Acknowledgments

This work was supported by the National Natural Science Foundation of China

(51402196), the Guangxi Science Foundation (Grants 2016GXNSFCB380006), and the Scientific Research Foundation of Guangxi University (Grant XTZ160530).

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Figures and Caption

Figure 1. a) XRD patterns of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0, 0.03, 0.06, 0.09$) ceramics, b) Enlarged (111) and (002) profiles of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0, 0.03, 0.06, 0.09$).

Figure 2. The surface SEM images of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics after thermal etched. a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

Figure 3. The $\varepsilon(T)$ and $\tan \delta(T)$ of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics under different frequencies and the Lorentz fitting of the $\varepsilon(T)$ at 100 Hz (dot line). a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

Figure 4. P - E curves of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics at selected temperatures. a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

Figure 5. The $(\partial P/\partial T)_E(T)$ curves and $P(T)$ curves (inset) of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics at selected electric fields. a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

Figure 6. The $\Delta T(T)$ of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics at selected electric fields. a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

Figure 7. The $\Delta S(T)$ of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ceramics at selected electric fields. a) $x = 0$, b) $x = 0.03$, c) $x = 0.06$ and d) $x = 0.09$.

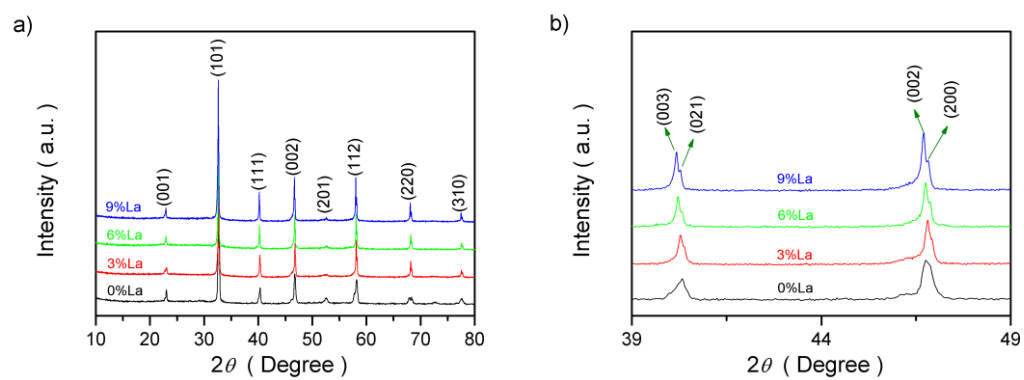


Figure 1

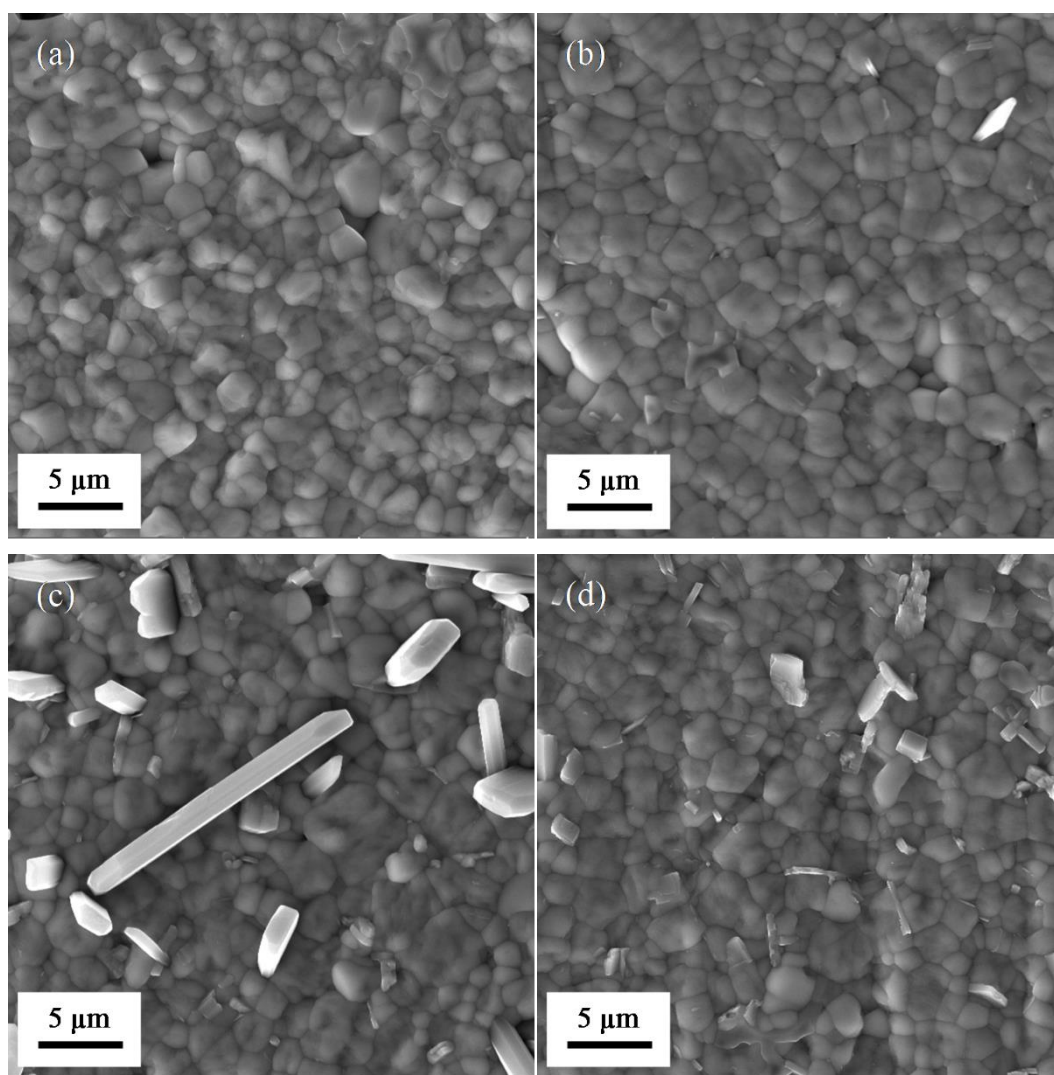


Figure 2

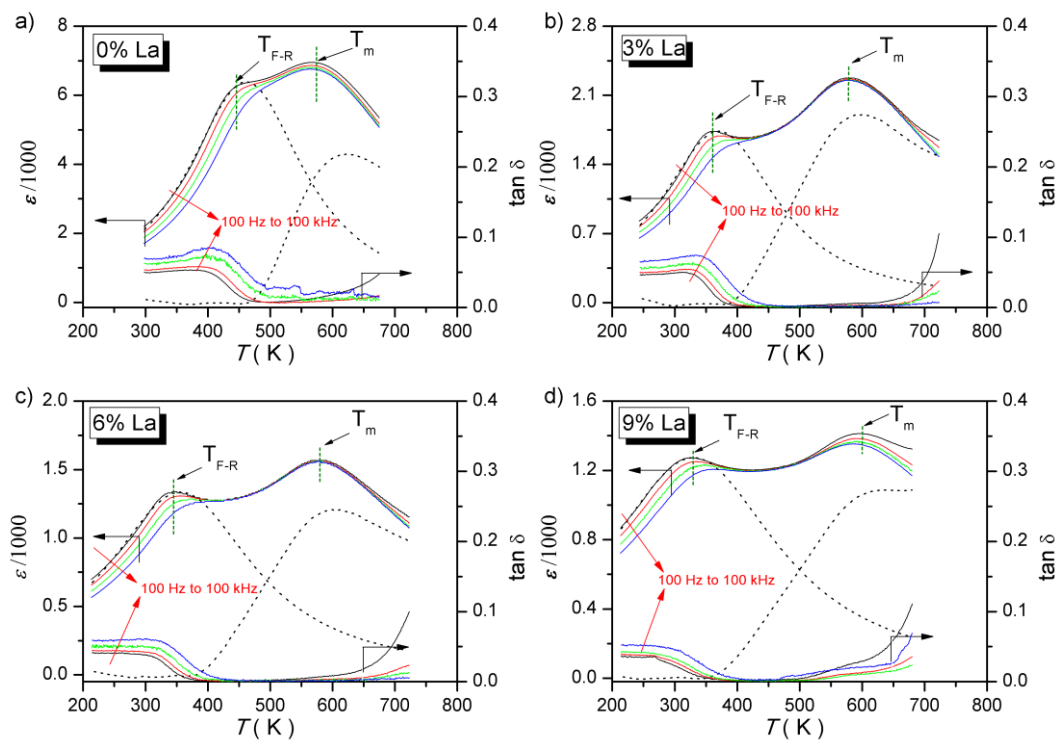


Figure 3

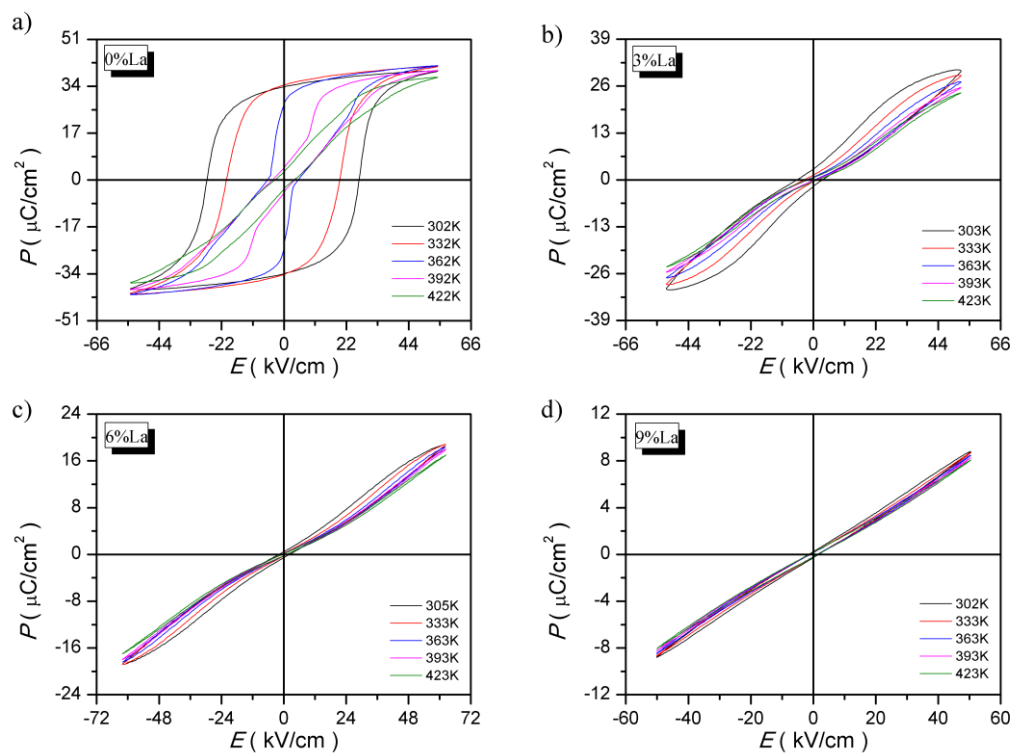


Figure 4

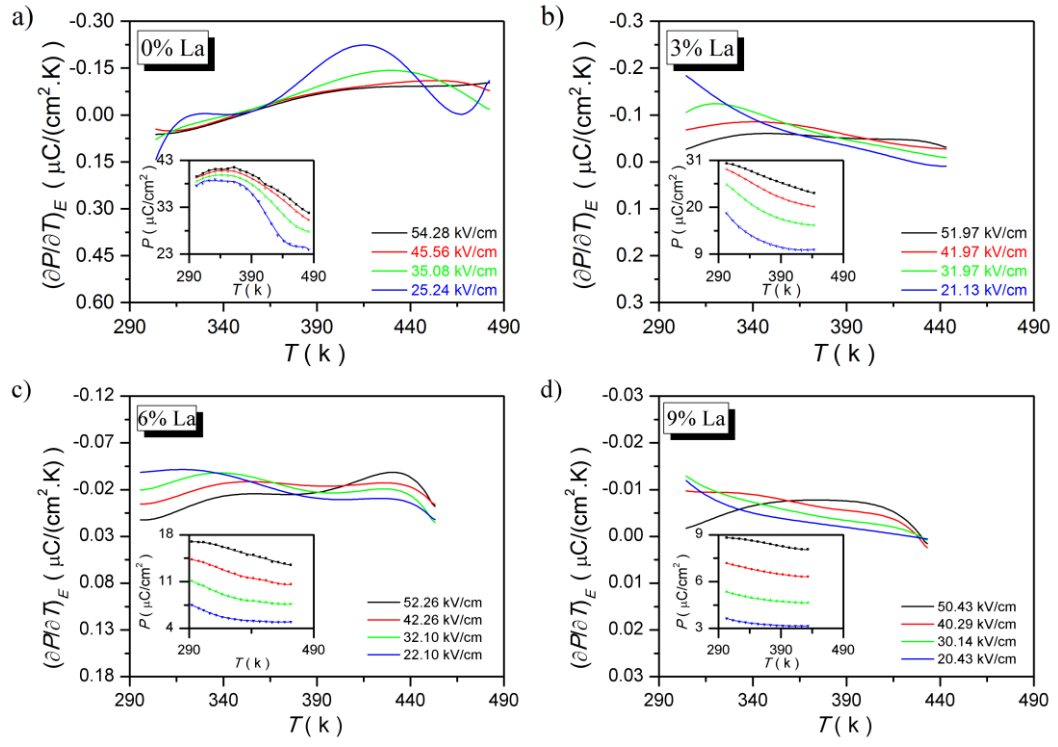


Figure 5

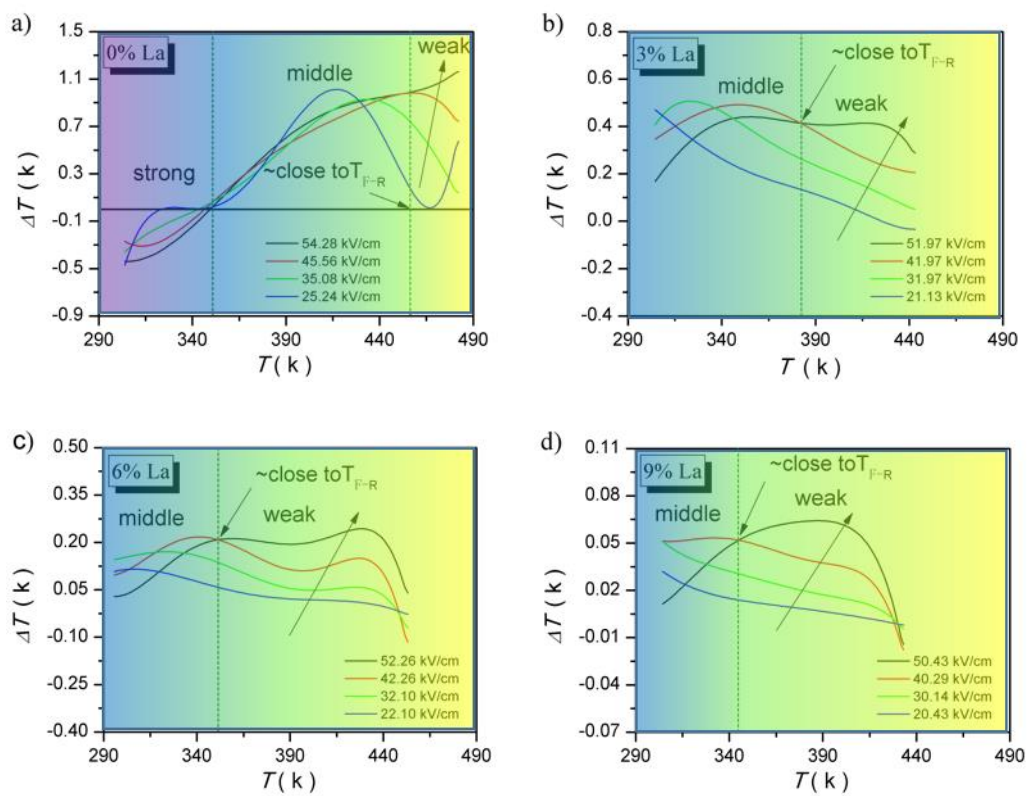


Figure 6

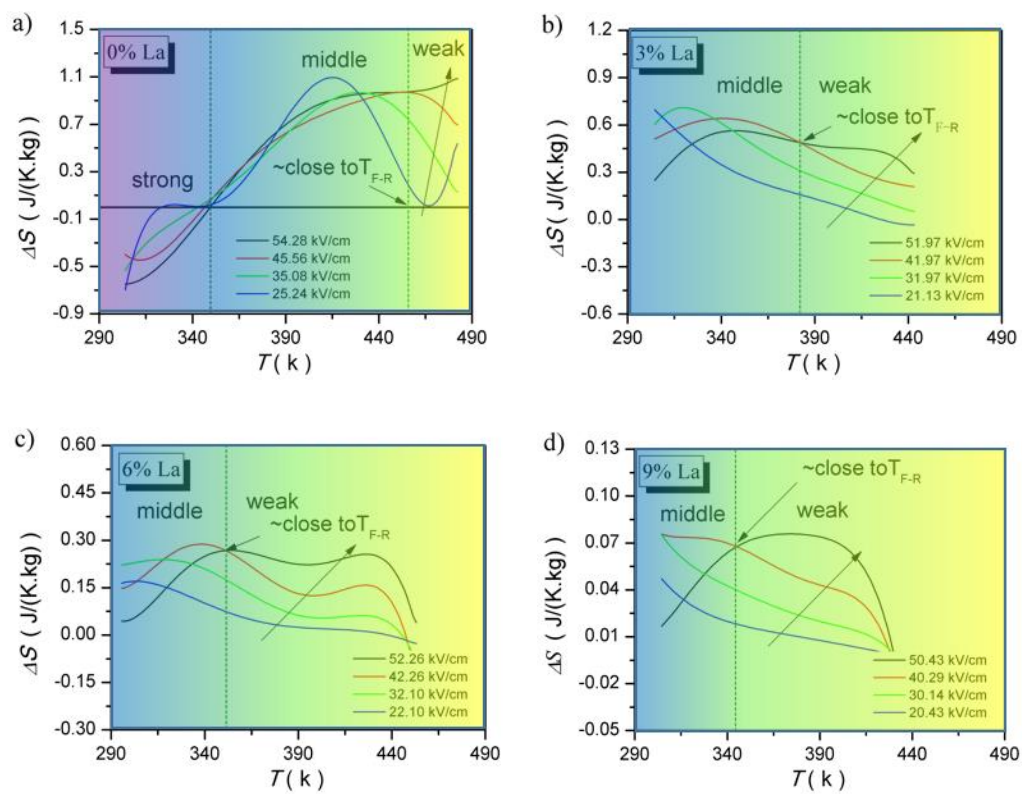


Figure 7

Table 1. Lorentz fitting results of the temperature dependence of the $\varepsilon(T)$ of $[(\text{Bi}_{1/2}\text{Na}_{1/2})_{0.94}\text{Ba}_{0.06}]_{(1-1.5x)}\text{La}_x\text{TiO}_3$ ($x = 0, 0.03, 0.06, 0.09$) ceramics at 100 Hz.

Sample	$T_A(\text{K})$	ε_A	$\delta_A(\text{K})$	$T_m(\text{K})$	ε_m
0.0	461	6.38	80	570	6.95
0.03	370	1.74	82	574	2.27
0.06	356	1.34	101	580	1.56
0.09	330	1.27	118	585	1.41



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2017-09-22

Elsevier

<http://dx.doi.org/10.1016/j.ceramint.2017.09.179>

Li L, Xu M, Zhang Q, et al., Electrocaloric effect in La-doped BNT-6BT relaxor ferroelectric ceramics. *Ceramics International*, Volume 44, Issue 1, January 2018, Pages 343-350